

ARKIV FÖR KEMI Band 1 nr 54

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Mixed bismuth oxides with layer lattices

I. The structure type of $\text{CaNb}_2\text{Bi}_2\text{O}_9$

By BENGT AURIVILLIUS

With 5 figures in the text

In the course of a comprehensive investigation of mixed bismuth oxides, the system Bi_2O_3 — TiO_2 was studied. At about 40 atomic % of TiO_2 a phase with a body-centered pseudo-tetragonal unit cell with $a = 3.84$ and $c = 32.8$ Å was found. X-ray analysis (to be published later) seemed to show that the structure was built up of $\text{Bi}_2\text{O}_2^{2+}$ layers parallel to the basal plane, and sheets of composition $\text{Bi}_2\text{Ti}_3\text{O}_{10}^{2-}$. The atomic arrangement within the $\text{Bi}_2\text{Ti}_3\text{O}_{10}^{2-}$ sheets seemed to be the same as in structures of the perovskite type and the structure could then be described as consisting of $\text{Bi}_2\text{O}_2^{2+}$ layers between which double perovskite layers are inserted.

An attempt was then made to synthesize compounds where the $\text{Bi}_2\text{O}_2^{2+}$ layers alternate with single perovskite layers. The general formula for such compounds could be expected to be: $(\text{MeBi})_6\text{R}_4\text{O}_{18}$. In actual fact compounds with this structure could be prepared with Me: Na K Ca Ba Sr Pb R: Ti Nb Ta.

Procedure: Weighed amounts of the appropriate oxides or carbonates were mixed and heated in platinum or gold crucibles to about 1000°C . A number of compounds with the general formula $(\text{Bi, Me})_6\text{R}_4\text{O}_{18}$ were prepared. Out of these the following were found to have a body-centered tetragonal or pseudo-tetragonal unit cell. The real unit cells, however, appeared to be face-centered-orthorhombic.

Composition	Orthorhombic description			Pseudo-tetragonal description	
	a	b	c	a	c
$\text{Bi}_3\text{NbTiO}_9$	5.405	5.442	25.11	3.836	25.11
$\text{Bi}_3\text{TaTiO}_9$	5.402	5.436	25.15	3.832	25.15
$\text{CaBi}_2\text{Nb}_2\text{O}_9$	5.435	5.485	24.87	3.860	24.87
$\text{SrBi}_2\text{Nb}_2\text{O}_9$	5.504	5.504	25.05	3.892	25.05
$\text{SrBi}_2\text{Ta}_2\text{O}_9$	5.509	5.509	25.06	3.895	25.06
$\text{BaBi}_2\text{Nb}_2\text{O}_9$	5.533	5.533	25.55	3.912	25.55
$\text{PbBi}_2\text{Nb}_2\text{O}_9$	5.492	5.503	25.53	3.887	25.53
$\text{KBi}_2\text{Nb}_2\text{O}_{18}$	5.506	5.506	25.26	3.893	25.26
$\text{NaBi}_2\text{Nb}_2\text{O}_{18}$	5.47	5.47	26.94	3.87	26.94

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Single crystals were prepared from the $\text{PbBi}_2\text{Nb}_2\text{O}_9$ and $\text{Bi}_3\text{NbTiO}_9$ phases. Weissenberg photographs of $0kl$ and $1kl$ (pseudo-tetragonal cell) were taken. In the powder photographs of $\text{Bi}_3\text{NbTiO}_9$ (Table 6a) the reflections 110 , 211 , 215 , 220 and 310 were clearly split up. No cleavage was found for the reflections $10l$, $20l$, and $30l$, (in all cases pseudo-tetragonal indices). From this it was concluded that the structure might be described by means of orthorhombic unit cells, having the same c axes as the pseudo-tetragonal cells, and a and b axes equal to the diagonals of the pseudo-tetragonal cells. Using orthorhombic units the Weissenberg photographs register hkl and $h, h+2, l$. In Table 6a the $\sin^2 \theta$ of $\text{Bi}_3\text{NbTiO}_9$ are calculated on the assumption of an orthorhombic unit cell.

A few discrepancies occur between the intensities of the spots as found in the Weissenberg photographs (first layer) and in the powder photographs. In the Weissenberg photographs ($h, h+2, l$), no difference was found between reflections hkl and hhl . From the powder photographs it is seen that 024 might be <204 and that $311 < 131$ and $3111 < 1311$. The reason might be the orientation of the powder.

 $\text{PbBi}_2\text{Nb}_2\text{O}_9$ phase

The powder photographs of $\text{PbBi}_2\text{Nb}_2\text{O}_9$ (Table 6b) could be explained assuming a tetragonal cell with $a = 3.887 \text{ \AA}$ and $c = 25.53 \text{ \AA}$, but for two lines being split up, which indicated an orthorhombic unit cell with axes $a = 5.492$, $b = 5.503$ and $c = 25.53 \text{ \AA}$. As in $\text{Bi}_3\text{NbTiO}_9$, it was thus assumed that the real symmetry is $D_{2h} - mmm$, though nothing in the Weissenberg photographs indicated a lower Laue symmetry than $D_{2h} - 4/mmm$. The observed density was 7.91 , thus allowing 4 formula units/unit cell ($d_{\text{calc.}} = 8.22$).

With the exception of the criterion for face-centering that hkl occurring only with h, k, l all odd or all even, no systematic extinctions were found. This is characteristic of the space groups D_{2h}^{23} , D_2^7 and C_{2v}^{18} .

Positions of the metal atoms

As the scattering factors for the Pb and the Bi atoms are almost the same, it makes no difference in the intensity calculations whether the Pb and the Bi atoms occupy separate positions or are mixed at random. Therefore no difference will be made between Pb and Bi; they will both be denoted by Me.

The intensities of the reflections seemed to depend mainly on the value of l (see Table 1a). It therefore seemed probable that at least the Me and the Nb atoms are placed along the lines: $(000; \frac{1}{2}\frac{1}{2}0; \frac{1}{2}0\frac{1}{2}; 0\frac{1}{2}\frac{1}{2}) + 00z$. The sum of $\sum I_{00l} \cos 2\pi lz$ and $\sum I_{11l} \cos 2\pi lz$ will under such conditions represent the Patterson function along $00z$. In Fig. 1a these sums are plotted as functions of z . It is seen from the graph that high maxima occur for $z = 0.20$ and $z = 0.40$.

The unit cell of $\text{PbBi}_2\text{Nb}_2\text{O}_9$ contains 12 Me atoms and 8 Nb atoms.

If the space groups are assumed to be D_{2h}^{23} , D_2^7 or C_{2v}^{18} a or b , the only way of placing 12 Me atoms on the lines $00z$ is in one 4-fold and one 8-fold position. With these assumptions the only 4-fold positions possible are 000 or $00\frac{1}{2}$.

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The crystals in the vicinity of maximum tetragonal indices $10l, 11l, 20l$ have been der

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Table 1a

Weissenberg Photographs of $\text{PbBi}_2\text{Nb}_2\text{O}_9$. Cu K_α radiation

The crystals form very thin plates, and therefore considerable absorption occurs. Spots in the vicinity of the lines described by WELLS (2) will therefore be weakened. The regions of maximum absorption are denoted by dotted lines. In the tables 1, 2 and 3 pseudo-tetragonal indices are used, and observed and calculated intensities for the reflections: 001, 101, 111, 201, and 211 are given. With orthorhombic description these reflections would have been denoted by: 001, 021 or 201, 221, 131 or 311.

l	I_1	I_2	I_{001}	$I_{\text{obs.}} I_{111}$	I_{201}
2	18	1.2		vw	vw
4	9.0	34	m^-	m	w
6	0.1	8.4	vvw	m	—
8	25	14	m^-	w	w^-
10	350	340	vst	m^+	m
12	19	0.2	w	—	—
14	36	100	m^+	vvw	w^-
16	9.6	4.8	vw	—	—
18	45	31	m	w	w
20	280	230	st	m^+	m^+
22	22	0.04	—	—	—
24	71	200	m^+	m	m^+
26	27	14	w	vvw	vw
28	85	26	m^-	w	w
30	190	130	m	m	
32	25	8.4	w		

l	I_1	I_2	I_{101}	$I_{\text{obs.}} I_{211}$	
1	1.7	17	m^-	—	
3	20	6.3	m^-	vvw	
5	380	370	vst	st	
7	18	0.0	—	—	
9	12	63	m	w	
11	3.2	5.3	—	—	
13	32	23	w	vvw	
15	320	280	st	w	
17	20	0.2	—	—	
19	55	160	m^+	w^+	
21	18	7.8	vvw	—	
23	61	32	m	w^+	
25	230	170	m^+	m	
27	24	1.7	—	—	
29	85	250	m^+		
31	48	23	w		

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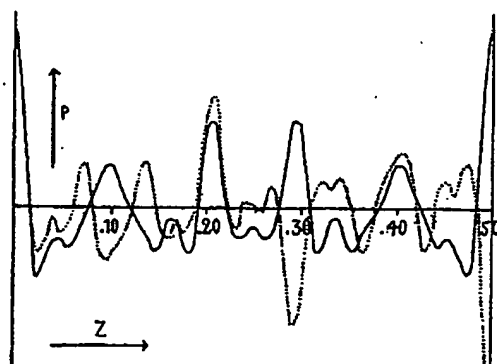
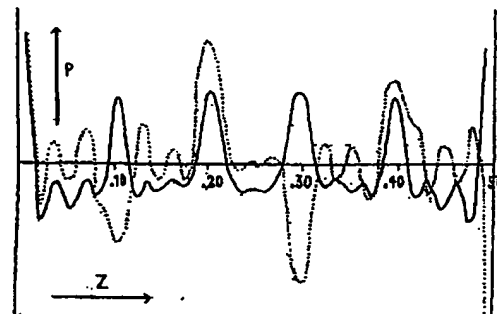
Table 1b
Weissenberg Photographs of $\text{PbBi}_2\text{Nb}_2\text{O}_9$

Zero layer				First layer			
001	I'_1	$I_{\text{obs.}}$	I'_2	211	I'_1	$I_{\text{obs.}}$	I'_2
4	11	m ⁻	32	1	1.7	—	14
6	1.4	vw	1.4	3	18	vw	4.8
8	28	m ⁻	14	5	330	st	340
10	360	vst	360	7	18	—	0.4
12	45	w	11	9	23	w	67
14	50	m ⁺	88	11	4.8	—	3.6
16	6.3	vw	5.8	13	26	vw	23
18	70	m	38	15	300	w	280
20	280	st	230	17	13	—	0.3
22	25	—	0.4	19	42	w ⁺	150
24	58	m ⁺	250	21	14	—	7.8
26	21	w	10	23	66	w ⁺	36
28	85	m ⁻	21	25	210	m	170
30	240	m	120	27	23	—	2.9
32	17	w	6.8				
101	I'_1	$I_{\text{obs.}}$	I'_2	111	I'_1	$I_{\text{obs.}}$	I'_2
1	1.0	m ⁻	8.4	2	18	vw	1.2
3	14	m ⁻	4.0	4	9.0	m	34
5	320	vst	340	6	0.1	m	8.4
7	16	—	0.5	8	25	w	14
9	24	m	67	10	350	m ⁺	340
11	6.3	—	2.0	12	19	—	0.2
13	24	w	24	14	36	vw	100
15	310	st	280	16	9.6	—	4.8
17	13	—	0.5	18	45	w	31
19	42	m ⁺	160	20	280	m ⁺	230
21	14	vw	7.8	22	22	—	0.0
23	64	m	38	24	71	m	200
25	210	m	170	26	27	vw	14
27	23	—	2.9	28	85	w	26
29	81	m ⁺	220	30	190	m	130
31	32	w	24				
201	I'_1	$I_{\text{obs.}}$	I'_2				
2	12	vw	1.4				
4	10	w	36				
6	0.5	—	4.8				
8	31	w ⁻	14				
10	360	m	350				
12	41	—	7.8				
14	50	w ⁻	88				
16	6.3	—	6.3				
18	69	w	40				
20	280	m ⁺	230				
22	25	—	0.4				
24	59	m ⁺	240				
26	20	vw	11				
28	85	w	21				

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Figure 1 a. Patterson function of $\text{PbV}_3\text{Bi}_2\text{Nb}_2\text{O}_9$ along $00z$ Full curve: $\sum_i I_{00i} \cos 2\pi iz$ Dotted curve: $\sum_i I_{11i} \cos 2\pi iz$ (orthorhombic indices).Figure 1 b. Patterson function of $\text{Bi}_3\text{NbTiO}_6$ along $00z$

Of 8-fold positions only $\pm 00z$ is possible.

It was assumed arbitrarily that 4 Me occupy the position 000.

Using the distances found with the aid of the Patterson function, two possibilities for placing the Nb atoms arose:

- | | | |
|---------------------------|-------------|------------------------|
| 1. 8 Nb in $\pm 00\ 0.20$ | 4 Me in 000 | 8 Me in $\pm 00\ 0.40$ |
| 2. 8 Nb in $\pm 00\ 0.40$ | 4 Me in 000 | 8 Me in $\pm 00\ 0.20$ |

The two curves on the graph were added and the areas under the peaks at 0.20 and 0.40 calculated. The ratio 1.5:1 was found for 0.20/0.40.

In case 1, the ratio was calculated to be 0.91:1 and in case 2, 1.1:1 if the ratio $f_{\text{Nb}}/f_{\text{Bi}}$ was assumed to be 0.46. These figures cannot be compared directly with the observed ratio 1.5:1 since the zero level in figure 1 is of course uncertain. Case 2 agrees slightly better insofar as the peak at 0.20 is actually higher. It seemed, however, that the uncertainty in determining the areas was so large that case 1 could not be excluded by these measurements alone.

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Case 1

In calculating the structure amplitudes z_{Nb} was varied around 0.20 and z_{Me} around 0.40. The average ratio f_{Nb}/f_{Me} was assumed to be 0.46. The intensities were compared with calculated values of A^2 :

$A = 10 (\cos 2\pi lz_{Me} + 0.46 \cos 2\pi lz_{Nb} + 0.5) = 10 F/4 f_{Me}$. In this way the best values for the parameters were found to be:

$z_{Me} = 0.397 \pm 0.002$ and $z_{Nb} = 0.192 \pm 0.004$. In Table 1a $I_{calc.}$ is compared with the observed intensities.

Case 2

z_{Nb} was varied about 0.40 and z_{Me} around 0.20. The best values were found to be $z_{Nb} = 0.412 \pm 0.004$ and $z_{Me} = 0.202 \pm 0.002$.

The observed and calculated values are compared in Table 1a.

It was found that arrangement 2 accounted slightly better for the experimental data than 1. It must, however, be borne in mind that the intensity ratios of weak spots might be changed through the influence of the oxygen atoms and that this influence was neglected in the calculations. The differences did not seem to be as large as to allow a decision between 1 and 2. It was therefore tried to find possible oxygen positions both for 1 and 2. The results were then compared.

Case 1. Positions of the oxygen atoms

The positions of the metal atoms were assumed to be: $(000; 0\frac{1}{2}\frac{1}{2}; \frac{1}{2}0\frac{1}{2}; \frac{1}{2}\frac{1}{2}0) + 000$ (4 Me_1) ± 00 0.397 (8 Me_2) ± 00 0.192 (8 Nb). Since all point positions of D_{2d}^{23} can be described by positions of D_2^7 or $C_{2v}^{18} a$, only D_2^7 and $C_{2v}^{18} a$ have been considered.

At first only D_2^7 will be discussed. If the interatomic distances O—O, Me—O and Nb—O should not be smaller than 2.5, 2.2 and 1.8 Å. oxygen atoms could only be situated in the following positions:

$$\begin{array}{llll}
 4(b) 00\frac{1}{2} & 4(c) \frac{1}{2}\frac{1}{2}\frac{1}{2} & 4(d) \frac{1}{2}\frac{1}{2}\frac{1}{2} & 8(g) \pm 00z \\
 8(h) \frac{1}{2}\frac{1}{2}z; \frac{1}{2}\frac{1}{2}\frac{1}{2}-z & & & 0.086 \leq z \leq 0.122 \\
 & 0.039 \leq |z| \leq 0.061 & & 0.262 \leq z \leq 0.311 \\
 & 0.148 \leq |z| \leq 0.201 & & \\
 8(i) \frac{1}{2}y\frac{1}{2}; \frac{1}{2}\frac{1}{2}-y\frac{1}{2} & 8(j) x\frac{1}{2}\frac{1}{2}; \frac{1}{2}-x\frac{1}{2}\frac{1}{2} & & \\
 y=0 & x=0 & &
 \end{array}$$

An attempt was made to find positions for the oxygen atoms giving approximately regular octahedra around Nb, since from known structures containing Nb^{5+} and O^{2-} this seemed to be the normal configuration $Nb^{5+}-O^{2-}$. The maximum distance of contact Nb—O was assumed to be 2.5 Å.

With these assumptions 8(g) and 8(h) are the only positions where oxygen atoms in contact with Nb can be situated.

With oxygen atoms in three 8-fold positions 8(h) the distances O—O would be too short. It then only remains to consider the case of oxygen atoms in two 8-fold positions 8(g) and two 8 fold positions 8(h). For the oxygen

atoms in contact with Nb, reasonable interatomic distances were obtained assuming: 8 O₂ in 8(g) $z_2 = 0.100$ 8 O₃ in 8(g) $z_3 = 0.264$ 8 O₄ in 8(h) $z_4 = 0.168$ 8 O₅ in 8(h) $z_5 = -0.168$. Even if small adjustments of these parameters are admitted for the remaining 4 O there is room only in the position $00\frac{1}{2}$ (O₁).

With these assumptions the distances would be:

Me ₁ -4 O ₁ = 2.75	Me ₂ -4 O ₂ 2.75	Nb-O ₂ = 2.34
Me ₁ -2 O ₃ = 2.55	Me ₂ -2 O ₄ 2.55	Nb-2 O ₄ = 2.04
	Me ₂ -2 O ₅ 2.55	Nb-O ₅ = 2.04
O ₄ -4 O ₅ = 2.74		Nb-O ₃ = 1.84
O ₃ -2 O ₄ = 2.60		
O ₃ -2 O ₅ = 2.60		
O ₁ -2 O ₂ = 2.55		

It is seen that the positions given might equally well be described by D_{2h}²³ or if the pseudo-tetragonal unit cell ($a = 3.89$ $c = 25.53$ Å) is chosen by D_{4h}¹⁷.

As $a \sim b$ and the positions of the oxygen atoms must be chosen from space considerations, the discussion will be the same for C_{2v}¹⁸ as for C_{2v}¹⁸. For C_{2v}¹⁸ it is found that the oxygen atoms can be only in the planes $y = 0$ $y = 0.25 \pm 0.03$ $y = \frac{1}{2}$ and $y = 0.75 \pm 0.03$. For $y = 0$ or $\frac{1}{2}$, z must either be 0 or $\frac{1}{2}$ or lie between the limits $0.049 < |z| < 0.451$, otherwise the distance O-O will be < 2.5 Å. For $y = \frac{1}{4}$ or $\frac{3}{4}$, z must have the values 0, $\frac{1}{2}$, $\frac{1}{4}$ or $\frac{3}{4}$ or lie between the limits $0.049 < |z| < 0.201$ $0.299 < |z| < 0.451$.

In figure 2a sections of the unit cell are made for $y = 0$ and $y = \frac{1}{2}$. Possible regions with space group C_{2v}¹⁸ are denoted in the figure by shaded areas. For these areas the distances O-O ≥ 2.5 Me-O ≥ 2.2 and Nb-O ≥ 1.8 Å.

With space group C_{2v}¹⁸ it thus seems that no basically new atomic positions are obtained, although this symmetry allows the atoms to be slightly shifted from the positions of D_{2h}²³.

In Table 1 the intensities have been calculated from the parameters found and compared with the observed ones. (The calc. intensities are denoted by I₁). The mode of calculation is shown by the calculation of I₀₀₁. $I = A^2$.

$A = 10 (0.5 + \cos 2\pi lz_{Me} + (f_{Nb}/f_{Me}) \cos 2\pi lz_{Nb} + (f_o/f_{Me}) (0.5 + \cos 2\pi lz_2 + \cos 2\pi lz_3 + 2 \cos 2\pi lz_4))$. Since the ratios f_{Nb}/f_{Me} and f_o/f_{Me} vary with $\sin \theta/\lambda$ they were interpolated from values given in the *International Tables* (1).

Case 2. Positions of the oxygen atoms

The positions of the metal atoms were assumed to be: $(000; \frac{1}{2} \frac{1}{2} 0; 0 \frac{1}{2} \frac{1}{2}; \frac{1}{2} 0 \frac{1}{2}) + 000$ (4 Me₁) ± 00 0.202 (8 Me₂) ± 00 0.412 (8 Nb).

With D_{2h}²³ the following positions are available for the oxygen atoms 4(b) $00\frac{1}{2}$

8(g) $\pm 00z$	8(h) $\frac{1}{4} \frac{1}{4} z, \frac{1}{4} \frac{1}{4} \bar{z}$
$0.086 \leq z \leq 0.116$	$0.039 \leq z \leq 0.161$
$0.288 \leq z \leq 0.342$	
4(c) $\frac{1}{4} \frac{1}{4} \frac{1}{4}$	4(d) $\frac{1}{4} \frac{1}{4} \frac{3}{4}$
	16(k) $xyz, \bar{x}\bar{y}z, x\bar{y}\bar{z}, \bar{x}y\bar{z}$
	$x = 0$ $x = 0.25 \pm 0.03$
	$y = 0.25 \pm 0.03$ or $y = 0$
	$z \sim 0.135$.

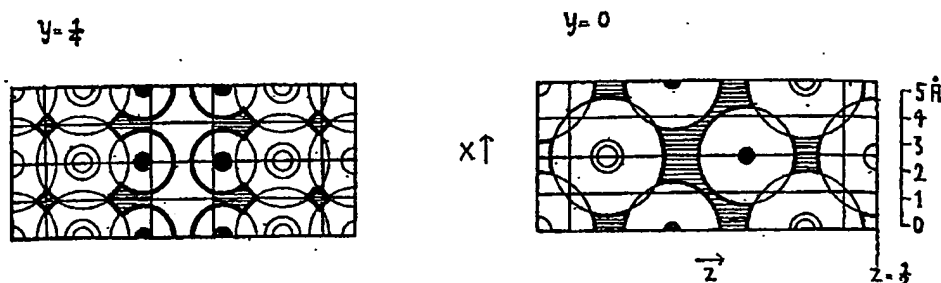
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Figure 2 a (see Case 1 in the text.)

The projection of the positions of the Nb, Me₁ and Me₂ atoms on the planes $y=0$ and $y = \frac{1}{2}$ are denoted by: black circles, white circles and double circles respectively.

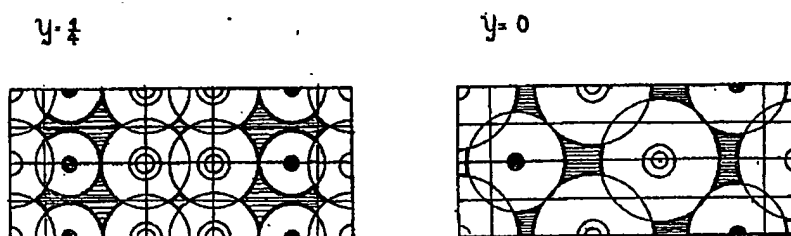
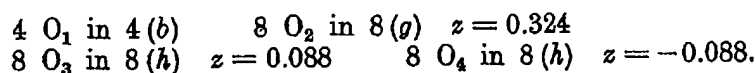


Figure 2 b (see Case 2 in the text.)

It was found that oxygen atoms in the positions 4(c), 4(d) or 16(k) could not be part of an octahedron around Nb. With O in the remaining positions 4(b), 8(g) and 8(h), octahedra around Nb might be achieved in the following ways: ($1.8 \leq \text{Nb}-\text{O} \leq 2.5$).

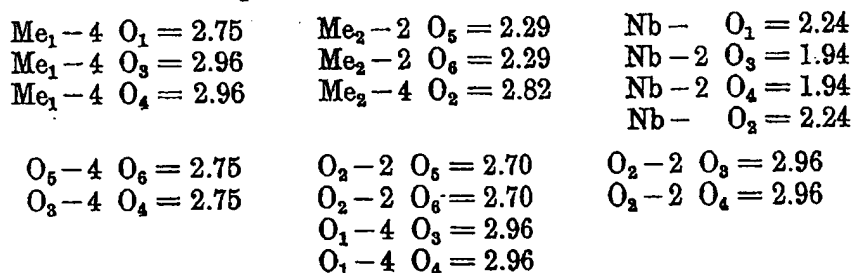
With oxygen atoms in three 8-fold positions 8(h), it seemed impossible to find positions for the remaining 12 oxygen atoms giving O—O distances ≤ 2.5 Å.

With two 8-fold positions 8(h) + 4(b) + one 8-fold position 8(g) the following positions were assumed for oxygen atoms in contact with Nb:



For the remaining 8 oxygen atoms there was only room in the positions 4(c) and 4(d). (O₅, O₆).

With the above assumptions the distances would be:



The above positions might be equally well described by D_{2h}²² or if a pseudo-tetragonal unit cell is assumed ($a = 3.89$ $c = 25.53$) by D_{4h}²³.

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In Figure 2b sections are made of the unit cell for $y = 0$ and $y = \frac{1}{4}$ showing the positions possible for the oxygen atoms if the space group C_{2v}^{18} is assumed. Possible regions are denoted by shaded areas. It was found that only with oxygen atoms situated near the positions given above, could octahedra of O around all Nb atoms and reasonable distances Nb-O be attained.

Thus no new arrangements were found when space group C_{2v}^{18} was assumed.

In Table 1 the intensities are calculated from the parameters given above. The calculated intensities are denoted by I_2 . The mode of calculation is the same as was used in case 1. It is seen from the Table that both 1 and 2, in which cases the influence of the oxygen atoms was neglected, account fairly well for the experimental data. From this follows that the calculated intensities I_1 and I_2 , where regard was taken to the O atoms, do not differ much either. Although the ratio 211:213 (see Table 1 (pseudo-tetragonal indices)) is best described by 1', 2' on the whole seemed to satisfy the observed intensities best (see for instance the intensity ratios 112:114 116:118 202:204 101:103 and 107:109). No definite conclusions could however be drawn from the study of $PbBi_2NbO_9$ alone.

 Bi_3NbTiO_9

Just as for $PbBi_2Nb_2O_9$, there was nothing in the Weissenberg photographs to indicate a lower Laue symmetry than $D_{4h}-4/mmm$. From the powder photographs (Table 6a) it is however seen that the actual unit cell is orthorhombic with axes $a = 5.405$ $b = 5.442$ $c = 25.11$ Å.

The intensities of the spots in the Weissenberg photographs indicate that the metal atoms are probably placed on the lines $00z$. The Patterson function (Fig. 1b) showed high maxima at 0.20 and 0.40. If the same assumptions are made as for $PbBi_2Nb_2O_9$, the following arrangements seemed to be possible:

A. 4 Nb in 000	B. 4 Ti in 000	C_1 4 Bi in 000
8 BiTi in $\pm 00z_1$	8 BiNb in $\pm 00z_1$	8 NbTi in $\pm 00z_1$
8 BiTi in $\pm 00z_2$	8 BiNb in $\pm 00z_2$	8 Bi in $\pm 00z_2$
C_2 4 Bi in 000		
8 NbTi in $\pm 00z_2$	$z_1 \sim 0.20$	$z_2 \sim 0.40$
8 Bi in $\pm 00z_1$		

The areas under the peaks at 0.20 and 0.40 were calculated as for $PbBi_2Nb_2O_9$ and the ratio 0.20/0.40 was found to be 1.4. The calculated ratios for A, B, C_1 and C_2 were 1.0, 1.0, 0.84 and 1.2. The area ratio for C_2 agreed best with the observed one. The differences are however small, so that all alternatives were considered. The intensities were calculated as for $PbBi_2Nb_2O_9$. The ratios f_{Ti}/f_{BiNb} , f_{Nb}/f_{BiTi} and f_{NbTi}/f_{Bi} were assumed to be 0.26, 0.57 and 0.34. The calculated and observed intensities for A and B are compared in Table 2. In these calculations the influence of the oxygen atoms was neglected. The best agreement was found for

	z_1	z_2
A	0.198	0.400
B	0.196	0.400

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Table 2
Weissenberg photographs of $\text{Bi}_3\text{NbTiO}_9$.

h	I_A	I_B	I_{00l}	$I_{\text{obs.}} I_{11l}$	I_{20l}
2	4.0	12	—	vw	vw
4	7.3	22	m	st	m ⁺
6	2.3	5.3	w	st	m
8	7.3	22	m	w	m
10	510	440	st	st	st
12	1.4	2.6	w	—	vw
14	15	52	m ⁺	vw	w
16	0.2	0.0	w	vw	—
18	11	30	m ⁺	w	vw
20	500	400	st	st	m ⁺
22	0.1	0.4	—	—	—
24	27	94	m	m	m
26	0.5	2.0	w	m	m
28	14	31	m	m ⁺	m ⁺
30	490	350	m	m ⁺	—
h	I_A	I_B	I_{10l}	I_{21l}	
1	4.4	12	st	m ⁺	
3	5.8	17	m	vw	
5	520	450	vst	vst	
7	2.6	6.3	—	—	
9	11	35	m	m	
11	0.8	1.4	w	w	
13	9.0	27	vw	w	
15	510	420	st	w	
17	0.6	0.3	—	—	
19	20	70	w	w	
21	0.0	0.5	w	w	
23	12	31	m	m	
25	500	380	m ⁺	m ⁺	
27	0.0	3.2	—	—	
29	34	120	m	—	
31	1.4	4.4	m	—	

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From Table 2 it is seen that A and B account quite well for the observed intensities.

With A and B, Bi and Ti or Bi and Nb would occupy the same point position. This seemed *a priori* unlikely and if it was assumed that $\text{Bi}_3\text{NbTiO}_9$ and $\text{PbBi}_2\text{Nb}_2\text{O}_9$ were built up in the same way, arrangements A and B would imply that Pb, Bi and Nb were distributed over one point position, in $\text{PbBi}_2\text{Nb}_2\text{O}_9$. Therefore, although arrangements A and B cannot be excluded from intensity discussions alone, they seem very improbable and will not be dealt with in the following.

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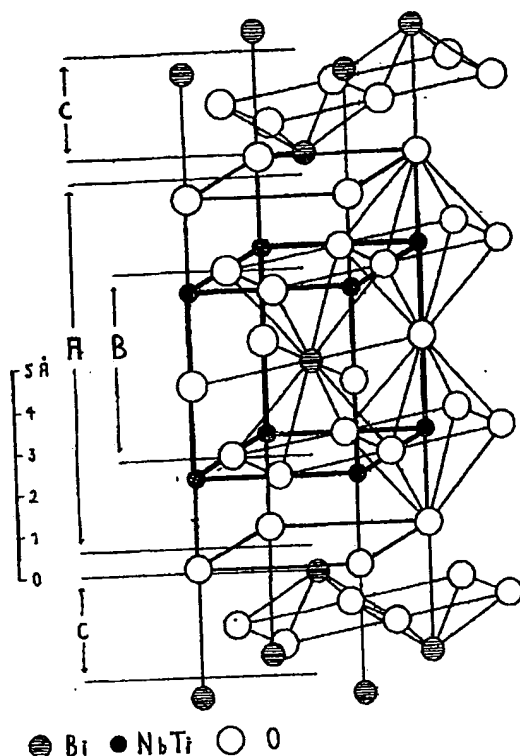


Figure 3.

One half of the pseudo-tetragonal unit cell of $\text{Bi}_4\text{NbTiO}_9$ (from $z \approx 0.25$ to $z \approx 0.75$). A denotes the perovskitic layer BiNbTiO_7 , C Bi_2O_3 layers and B the unit cell of a hypothetical perovskite structure $\text{BiNb}_{0.5}\text{Ti}_{0.5}\text{O}_3$.

Case C'_1

The following positions were assumed: 4 Bi in 000, 8 Bi in $\pm 00\ 0.396$, 8 NbTi in $\pm 00\ 0.192$. By the same arguments as used for $\text{PbBi}_2\text{Nb}_2\text{O}_9$, the following positions were arrived at:

$$\begin{array}{lll} 4\ \text{O}_1\ \text{in}\ 00\frac{1}{2} & 8\ \text{O}_2\ \text{in}\ \pm 00\ 0.092 & 8\ \text{O}_3\ \text{in}\ \pm 00\ 0.268 \\ 8\ \text{O}_4\ \text{in}\ \frac{1}{4}\frac{1}{4}z; \frac{1}{4}\frac{1}{4}\frac{1}{2}-z & z = 0.164 & \\ 8\ \text{O}_5\ \text{in}\ \frac{1}{4}\frac{1}{4}z; \frac{1}{4}\frac{1}{4}\frac{1}{2}-z & z = -0.164. & \end{array}$$

In Table 3 the intensities are calculated from these parameters.

Case C'_2

4 Bi in 000, 8 Bi in $\pm 00\ 0.200$ and 8 NbTi in $\pm 00\ 0.412$ were assumed. In case 2' (see $\text{PbBi}_2\text{Nb}_2\text{O}_9$) the parameters for the oxygen atoms would be 4 O_1 in $00\frac{1}{2}$, 8 O_2 in $\pm 00\ 0.324$, 4 O_5 in $\frac{1}{4}\frac{1}{4}\frac{1}{2}$.

$$\begin{array}{lll} 4\ \text{O}_6\ \text{in}\ \frac{1}{4}\frac{1}{4}\frac{1}{2}, & 8\ \text{O}_4\ \text{in}\ \frac{1}{4}\frac{1}{4}z; \frac{1}{4}\frac{1}{4}\frac{1}{2}-z & z = 0.088 \\ & 8\ \text{O}_3\ \text{in}\ \frac{1}{4}\frac{1}{4}z; \frac{1}{4}\frac{1}{4}\frac{1}{2}-z & z = -0.088 \end{array}$$

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Table 3
Weissenberg photographs of $\text{Bi}_2\text{NbTiO}_9$

Zero layer				First layer			
00l	$I_{C'_1}$	$I_{\text{obs.}}$	$I_{C'_2}$	21l	$I_{C'_1}$	$I_{\text{obs.}}$	$I_{C'_2}$
4	13	m	36	1	2.6	m ⁺	22
6	0.5	w	10	3	35	vw	6.3
8	48	m	16	5	290	vst	300
10	310	st	330	7	19	—	1.0
12	46	w	12	9	27	m	55
14	38	m ⁺	59	11	3.6	w	18
16	2.6	w	29	13	52	w	26
18	100	m ⁺	48	15	270	w	270
20	240	st	240	17	6.8	—	0.8
22	12	—	0.5	19	45	w	90
24	42	m	140	21	14	w	40
26	29	w	42	23	110	m	46
28	100	m	35	25	200	m ⁺	200
30	190	m	150				
10l	$I_{C'_1}$	$I_{\text{obs.}}$	$I_{C'_2}$	11l	$I_{C'_1}$	$I_{\text{obs.}}$	$I_{C'_2}$
1	2.0	st	17	2	34	vw	2.6
3	29	m	5.3	4	2.9	st	41
5	270	vst	300	6	6.8	st	69
7	17	—	1.2	8	62	w	22
9	29	m	56	10	360	st	310
11	5.8	w	15	12	14	—	1.0
13	49	vw	28	14	29	vw	76
15	270	st	270	16	10	vw	40
17	5.3	—	0.6	18	64	w	16
19	44	w	92	20	250	st	210
21	16	w	38	22	21	—	0.0
23	110	m	46	24	66	m	92
25	200	m ⁺	200	26	32	m	61
27	4.4	—	1.2	28	130	m ⁺	27
29	65	m	110	30	150	m ⁺	200
31	41	m	76				
20l	$I_{C'_1}$	$I_{\text{obs.}}$	$I_{C'_2}$				
2	18	vw	2.9				
4	12	m ⁺	37				
6	0.1	m	16				
8	48	m	17				
10	300	st	320				
12	33	vw	5.8				
14	37	w	58				
16	2.6	—	31				
18	100	vw	45				
20	240	m ⁺	240				
22	12	—	0.5				
24	42	m	140				
26	29	m	45				
28	100	m ⁺	35				

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In Table 3 the intensities are calculated with these assumptions.. It is seen from the Table that with C_1 the order of the reflections 101:103, 211:213, 202:204 and 107:109 are reversed. The same result was obtained if the influence of the oxygen atoms was neglected. If z_{Bi} and z_{NbTi} were varied around 0.396 and 0.192 so as to give correct ratios for some of these reflections, large discrepancies occurred for other reflections. With C_2 the intensities turned out to be of the right order. There are however a few discrepancies. 00 20 is certainly stronger than 0018 and $00\ 30 > 0028$, but the calculated ratios seem to be too large. On the whole the agreement is however good.

Thus if the X ray data for only one of $PbBi_2Nb_2O_9$ or Bi_3NbTiO_9 were considered, different atomic arrangements appeared to be possible, whereas only C_2 seems to explain the observed data both for $PbBi_2Nb_2O_9$ and Bi_3NbTiO_9 .

With orthorhombic description the positions will be:

$D_{2h}^{22} - mmm$

$(000; 0\frac{1}{2}\frac{1}{2}; \frac{1}{2}0\frac{1}{2}; \frac{1}{2}\frac{1}{2}0) +$

4 Bi_1 (Me_1) in 4 (a) 000
 8 Bi_2 (Me_2) in 8 (i) $\pm 00\ 0.200$ (0.202)
 8 NbTi (Nb) in 8 (i) $\pm 00\ 0.412$ (0.412)
 4 O_1 in 4 (b) $00\frac{1}{2}$
 8 O_2 in 8 (i) $00\ 0.324$ (0.324)
 8 O_3 in 8 (f) $\frac{1}{4}\frac{1}{4}\frac{1}{4}; \frac{1}{4}\frac{1}{4}\frac{3}{4}$
 16 O_4 in 16 (j) $\frac{1}{4}\frac{1}{4}z; \frac{1}{4}\frac{1}{4}\bar{z}; \frac{1}{4}\frac{3}{4}z; \frac{1}{4}\frac{3}{4}\bar{z}$
 $z = 0.088$.

With pseudo-tetragonal description the positions will be:

$D_{4h}^{17} - I\ 4/mmm$

$(000; \frac{1}{2}\frac{1}{2}\frac{1}{2}) +$

2 Bi_1 (Me_1) in 2 (a) 000
 4 Bi_2 (Me_2) in 4 (e) $\pm 00\ 0.200$ (0.202)
 4 NbTi (Nb) in 4 (e) $\pm 00\ 0.412$ (0.412)
 2 O_1 in 2 (b) $00\frac{1}{2}$
 4 O_2 in 4 (e) $\pm 00\ 0.324$ (0.324)
 4 O_3 in 4 (d) $0\frac{1}{2}\frac{1}{2}; \frac{1}{2}0\frac{1}{2}$
 8 O_4 in 8 (g) $\pm (0\frac{1}{2}z; \frac{1}{2}0z)$ $z = 0.088$.

Table 4

Values of the tolerance factor, t , for different compounds having the $CaBi_2Nb_2O_9$ structure.

Compound	$t \cdot 100$
Bi_3NbTiO_9	91
Bi_3TaTiO_9	91
$CaBi_2Nb_2O_9$	91
$SrBi_2Nb_2O_9$	99
$SrBi_2Ta_2O_9$	99
$BaBi_2Nb_2O_9$	106
$PbBi_2Nb_2O_9$	101
$NaBi_2Nb_2O_{18}$	91
$KBi_2Nb_2O_{18}$	97

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Table 5

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Powder photographs of $\text{CaBi}_2\text{Nb}_2\text{O}_9$ and $\text{SrBi}_2\text{Nb}_2\text{O}_9$. Cr K radiation. Pseudo-tetragonal indices.

$\text{CaBi}_2\text{Nb}_2\text{O}_9$				
hkl	$I_{\text{obs.}}$	I_{α}	I_{β}	I_{γ}
006	—	8.4	0.8	0.01
008	m	61	9.6	0.6
114+00 10	(m)	230	230	230
00 12	—	0.5	19	47
00 14	vw	29	46	56
00 16	vw	2.6	0.6	4.0
00 18	w	100	34	13
101	—	1.2	0.5	2.9
103	st	36	1.4	1.4
105	vst	200	200	200
107	w	21	0.3	7.8
109	vvw	18	32	43
10 11	—	7.0	0.2	0.5
10 13	m ⁺	79	19	4.4
20 10+10 15	(st)	160	160	160
21 11+10 17	(w)	22	0.2	9.0
112	—	26	0.04	4.8
00 10+114	(m)	4.4	14	22
116	w	4.4	17	26
118	m	69	13	1.7
204+11 10	(st)	200	200	200
11 12	—	22	0.1	8.4
11 14	w	41	61	72
202	w	27	0.1	4.4
11 10+204	(st)	3.6	13	20
206	w	3.2	0.04	1.4
215+208	(st)	59	9.0	0.5
10 15+20 10	(st)	220	220	220
20 12	vw	1.4	15	41
211	w ⁻	0.1	2.3	6.3
213	w	38	2.0	1.0
208+215	(st)	210	210	210
217	—	21	0.3	7.8
219	vvw	18	33	43
10 17+21 11	(w)	4.4	0.01	1.4
$\text{SrBi}_2\text{Nb}_2\text{O}_9$				
006	—	3.2	0.1	0.2
008	vw	45	11	2.6
114+00 10	(m)	260	260	260
00 12	vw	0.2	16	35
00 14	m	42	58	64
00 16	—	0.3	1.4	4.4
00 18	w	83	35	18
00 20	st	150	160	160

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 $\text{KBi}_2\text{Nb}_2\text{O}_9$
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Table 5 (cont.)

	<i>hkl</i>	<i>I</i> _{obs.}	<i>I</i> _α	<i>I</i> _β	<i>I</i> _γ
i. Pseudo	101	—	0.0	2.0	4.0
	103	vw	24	2.0	0.1
	105	st	230	240	230
	107	—	12	0.01	3.6
	109	w	28	42	49
	10 11	—	2.3	0.01	0.8
	10 13	m	61	20	8.4
	20 10+10 15	(st)	190	200	200
	β 00 20+220+10 17	(m)	13	0.01	4.0
	10 19	m	88	110	120
	112	—	16	0.3	1.7
	00 10+114	(m)	10	20	26
	116	vw	10	22	29
	118	w	52	14	4.4
	11 10	st	230	240	230
	11 12	—	13	0.0	3.6
	11 14	w	56	74	81
	11 16	—	0.1	4.0	8.4
	11 18	m	40	10	2.3
	202	—	17	0.4	1.4
	204	—	9.0	19	24
	206	—	0.5	0.6	2.3
	β 11 14+208	(w)	43	10	2.3
	10 15+20 10	(st)	250	260	250
	β 21 13+20 12	(vw)	0.01	12	29
	20 14	w	42	58	64
	20 16	—	0.2	1.7	4.8
	211	w	0.6	4.8	7.8
	213	w	26	2.6	0.01
	215	st	240	240	240
	217	—	12	0.01	3.6
	219	vw	27	42	49
	21 11	—	1.2	0.4	2.0
	21 13	m	59	19	7.8
	20 10+21 15	(st)	190	190	190

One half of the pseudo-tetragonal unit cell is pictured in figure 3. As mentioned in the discussion on $\text{PbBi}_2\text{Nb}_2\text{O}_9$, it was impossible in this phase to determine how Pb and Bi are distributed over the point positions 000 and 00 0.202. It therefore seemed of interest to try to determine the positions of Ca, Sr, Ba and K in the compounds $\text{CaBi}_2\text{Nb}_2\text{O}_9$, $\text{SrBi}_2\text{Nb}_2\text{O}_9$, $\text{BaBi}_2\text{Nb}_2\text{O}_9$ and $\text{KBi}_2\text{Nb}_2\text{O}_9$. As the cell dimensions of the Pb, Ca, Sr, Ba and K compounds do not differ much it was assumed that the parameters of $\text{PbBi}_2\text{Nb}_2\text{O}_9$ are also valid for the other compounds. There were three extreme ways of distributing Ca, Sr, Ba and K over the positions 000 and 00 0.202:

- α Only Bi in 00 202
- β Random distribution
- γ Only Bi in 000.

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Table 6 a

Powder photographs of $\text{Bi}_3\text{NbTiO}_9$. Cr K radiation.

Orthorhombic indices.

hkl	$10^4 \sin^2 \theta_{\text{calc.}}$	$10^4 \sin^2 \theta_{\text{obs.}}$	$I_{\text{obs.}}$	hkl	$10^4 \sin^2 \theta_{\text{calc.}}$	$10^4 \sin^2 \theta_{\text{obs.}}$	$I_{\text{obs.}}$	hkl	I
111	0914	0909	m	0214	5856	5874	m	101	
113	1080	1072	w+	2014	5878			103	
008	1333	1319	w+			5990	vw	008	
116	1414	1413	vst	139	6128	6129	w+	105	
020	1774	1767	m+	319	6172	6175	w+	110	
200	1796	1790	m			6250	vvw	0010	
0010	2083	2078	m	2212	6569	6573	vvw	114	
204	2129			0018	6747	6747	vw	β 109	
β 119	2136	2131	w	1311	6961	6961	w	116	
		2324	vvw	040	7097			109	
026	2524	2530	m	0216	7105	7097	m	β 200	
206	2546	2546	m	2016	7127			0012	
119	2580	2586	m	042	7180	7182	m	118	
		2739	vvw	400	7185			1011	
028	3107			β 240	7357			200	
208	3129	3129	w	β 420	7420	7388	w	1110	
1111	3413	3415	w	044	7430			0014	
		3523	w	2214	7652	7648	w	1013	
220	3570	3576	st	046	7847	7836	vw	211	
222	3653			406	7935			208	
β 1113	3654	3648	vw	1313	7960	8005	w	215	
0210	3857			3113	8004			1015	
2010	3879	3860	st	331	8054			2010	
0014	4082			0020	8330	8330	m		
β 135	4103	4089	w	1119	8411	8409	m		
		4236	vw	048	8430				
226	4320	4324	w	2018	8543	8558	st		
1113	4412	4414	w	335	8554				
131	4462	4454	w	240	8887	8888	m		
311	4506	4504	vw	2216	8901				
β 1115	4621			420	8959	8963	m		
133	4628	4623	w	242	8970				
313	4672			1315	9127	9105	st		
β 2210	4679	4676	w	3115	9171	9156	vst		
228	4903	4896	w	0410	9180	9189	m		
135	4962	4956	st	4010	9268	9264	m		
315	5006	5006	st	424	9292				
				426	9709	9712	w		
1115	5579	5581	st	339	9720				
2210	5653	5652	st						

The intensities were calculated for these possibilities by calculations similar to those for $\text{PbBi}_2\text{Nb}_2\text{O}_9$, and compared with the observed ones. See Table 5. It was found that in no case did γ explain the observed intensities. For the Sr, Ba and K compounds the observed intensities did not permit any decision between α and β . For $\text{CaBi}_2\text{Nb}_2\text{O}_9$, however, only α seemed to give correct intensities. It was therefore concluded that the compounds discussed have the α arrangement.

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Table 6b

Powder photographs of $\text{PbBi}_2\text{Nb}_2\text{O}_9$. Cr K radiation.

Pseudo-tetragonal indices.

$I_{\text{obs.}}$	hkl	$10^4 \sin^2 \theta_{\text{calc.}}$	$10^4 \sin^2 \theta_{\text{obs.}}$	$I_{\text{obs.}}$	hkl	$10^4 \sin^2 \theta_{\text{calc.}}$	$10^4 \sin^2 \theta_{\text{obs.}}$	$I_{\text{obs.}}$
m	101	0889	0892	vw	1114	5682	5699	m
vw	103	1050	1052	vvw	219	5974	5990	m
w+	008	1288	1285	vvw	0018	6520	6529	w
w+	105	1372	1371	vst	0020	6663		
vw	110	1738	1753	m	1017	6685	6664	vvw
vw	0010	2012	2019	m	1116	6889		
vw	114	2060			β 305	6903	6924	m
w	β 109	2089	2068	vvw	220	6950	6934	m
m	116	2462	2469	m	2014	7419	7435	m
m	109	2499	2499	m	2113	7744	7749	m
m	β 200	2877			301	7839	7846	vw
m	0012	2898	2881	m	0020	8049	8052	m
w	118	3026	3028	w	1019	8133	8131	m+
vw	1011	3304	3300	vvw	228	8238		
w	200	3475	3475	st	1118	8258	8268	vw
w	1110	3750	3755	st	305	8332	8339	m+
w	0014	3944	3946	w+	310	8688	8673	m+
vw	1013	4269	4270	w+	2115	8872	8690	m+
w	211	4364	4355	vvw	2210	8962	8874	st
m	208	4763	4770	vvw	314	9010	8980	m
m	215	4847	4850	st	309	9449	9450	vvw
st	1015	5397	5416	st				
m	2010	5487	5493	st				

The structure of $\text{Bi}_3\text{NbTiO}_9$ is thus built up of $\text{Bi}_2\text{O}_7^{2+}$ layers between which BiNbTiO_7^- layers are inserted. The structure may be looked upon as a perovskite structure where perovskite layers are separated by $\text{Bi}_2\text{O}_7^{2+}$ layers. This view was supported by the fact that in all cases where the above structure was observed the radii of the ions in the layers lying between the $\text{Bi}_2\text{O}_7^{2+}$ layers would allow for the formation of a perovskite structure. If the tolerance factor t is calculated from the ionic radii of the elements constituting the layers between the $\text{Bi}_2\text{O}_7^{2+}$ layers, it is found to lie between the limits 0.9 and 1.1 (see Table 4), the same limits within which perovskite structures are found to be stable.

t was calculated from the formula: $1.06 (R_A + R_O) = 0.95 t \sqrt{2} (R_B + R_O)$. (See (3).)

$A = (\text{K} + \text{Bi})/2$ Ca, Sr, Ba etc.

$B = (\text{Nb} + \text{Ti})/2$ (Ta + Ti)/2 Nb. Ta.

In the calculations case α was assumed. For calculating t the following values for the ionic radii were used: Bi^{3+} 1.00, Ba^{2+} 1.39, Sr^{2+} 1.20, Ca^{2+} 1.02, Pb^{2+} 1.26, Na^+ 0.97, K^+ 1.33, Nb^{5+} 0.69, Ta^{5+} 0.69, Ti^{4+} 0.66 and O^{2-} 1.36. If, for instance, 1 K + 1 Bi are distributed over one 2-fold position the radius of (K, Bi) was taken as $(r_K + r_{\text{Bi}})/2$.

B. AURIVILLIUS, *Mixed bismuth oxides with layer lattices*

The structure proposed for $\text{CaBi}_2\text{Nb}_2\text{O}_9$ reminds one of the structure of beyerite $\text{CaBi}_2\text{O}_3(\text{CO}_3)_2$ described by LAGERCRANTZ and SILLÉN (4). The unit cell of beyerite is body-centered tetragonal with $a = 3.767$ and $c = 21.690 \text{ \AA}$. The positions of the Ca and the Bi atoms in beyerite are Ca 000 and $\text{Bi} \pm 00 \ 0.19$ (space group D_{2h}^{17}), as compared with $a = 3.860$ $c = 24.87$, Ca in 000 and Bi in $\pm 00 \ 0.20$ for $\text{CaBi}_2\text{Nb}_2\text{O}_9$.

The "rotating" CO_3^{2-} groups in beyerite correspond to octahedral sheets $\text{Nb}_4\text{O}_{12}^{2-}$ in $\text{CaBi}_2\text{Nb}_2\text{O}_9$. Following the notations given by LAGERCRANTZ and SILLÉN (4) the structure described above might also be denoted by X_{11} .

SUMMARY

A series of tetragonal or pseudo-tetragonal phases of general composition $(\text{Bi}, \text{Me})_2\text{R}_2\text{O}_{18}$ have been investigated.

Me: Na, K, Ca, Sr, Ba, Pb.

R: Nb, Ta, Ti.

The positions of the Me and R atoms were determined from the observed intensities and the positions of the O atoms were deduced from space considerations.

The following structure is proposed:

$D_{2h}^{25} - F mmm$

(000; $0 \frac{1}{2} \frac{1}{2}$; $\frac{1}{2} 0 \frac{1}{2}$; $\frac{1}{2} \frac{1}{2} 0$) +
 4 Bi in 4 (a) 000
 8 Bi in 8 (i) $\pm 00 \ 0.200$
 8 NbTi in 8 (i) $\pm 00 \ 0.412$
 4 O in 4 (b) $00 \frac{1}{2}$
 8 O in 8 (i) $\pm 00 \ 0.324$
 8 O in 8 (f) $\frac{1}{4} \frac{1}{4} \frac{1}{4}$; $\frac{1}{4} \frac{1}{4} \frac{3}{4}$
 16 O in 16 (j) $\frac{1}{4} \frac{1}{4} z$; $\frac{1}{4} \frac{1}{4} \bar{z}$; $\frac{1}{4} \frac{3}{4} z$; $\frac{1}{4} \frac{3}{4} \bar{z}$ $z = 0.088$.

From intensity calculations it was found that Ca in $\text{CaBi}_2\text{Nb}_2\text{O}_9$ and Sr, Ba and K in the corresponding compounds are probably situated in the position 000. The proposed structure is built up of $\text{Bi}_2\text{O}_7^{2+}$ layers alternating with single perovskite layers. The resemblance to the structure of beyerite is pointed out.

I wish to thank Professor L. G. SILLÉN for valuable discussions concerning this work.

Stockholms Högskola, Institute of Inorganic and Physical Chemistry, June 1949.

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